

SOLID-STATE REACTIONS IN THE  
 $\text{SiO}_2\text{-Al}_2\text{O}_3$  BINARY SYSTEM

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Karlsruhe Univ., W. Germany**Abstract**

The solid-state reactions between  $\text{SiO}_2$  and  $\text{Al}_2\text{O}_3$  were studied at 1200°C up to 1550°C. Hot press technique was used to obtain very good contact between the reactants. The influence of pressure, time of reaction and the shape (powder or compact) and the form of solids (amorphous or crystalline) were studied and correlated. The reaction was either continued in the hot press conditions or annealed at different time. The thickness of the reaction product as well as its chemical composition was measured with scanning electron microprobe analyser. Surface area, pore size distribution and optical microscope were also used.

**Introduction**

The solid state reaction in the  $\text{SiO}_2\text{-Al}_2\text{O}_3$  binary system was studied by Bowen and Greig(1). They observed the presence of mullite. Although numerous investigators have challenged this classical diagram, its validity has never been in doubt(2). Mullite is a solid solution between  $\text{SiO}_2$  and  $\text{Al}_2\text{O}_3$  which can be separated from the molten state. The marginal composition of mullite is 3  $\text{Al}_2\text{O}_3$ , 2  $\text{SiO}_2$  ( $\text{A}_{35}\text{S}_2$ ) with 72% by weight of  $\text{Al}_2\text{O}_3$  while mullite 2  $\text{Al}_2\text{O}_3$ ,  $\text{SiO}_2$  ( $\text{A}_{25}$ ) contains 78%  $\text{Al}_2\text{O}_3$ . Most of studies in this system were carried out by either static method of quenching or by differential thermal analysis and recently by splat-cooling and flame-spraying which led to obtain amorphous material(3), while the alkoxide method of r.f sputtering was applied to obtain various amorphous products between  $\text{SiO}_2$  and  $\text{Al}_2\text{O}_3$  (4). Weisweiler and Serry(5) studied the reaction between  $\text{SiO}_2$  crystal and  $\text{Al}_2\text{O}_3$  powder by annealing the pre-hot pressed samples at 1000-1550°C for different times. They observed the presence of glass phases with mullite.

In the present investigation the kinetics and mechanisms of the solid-state reaction between  $\text{SiO}_2$  and  $\text{Al}_2\text{O}_3$  at 1200-1550°C, are discussed. The reaction at the interface between the reactants was examined by light microscope while the analysis of the products was determined by scanning electron microprobe analyser (SEMPA).



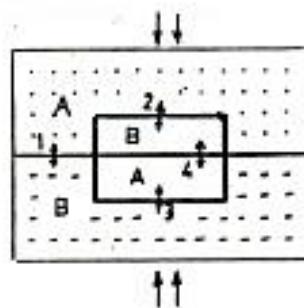


Fig. 1 Reaction Diagram in the  $\text{SiO}_2$ - $\text{Al}_2\text{O}_3$  Binary System.

- (1) Powder A- Powder B
- (2) Powder A- Compact B
- (3) Powder B- Compact A
- (4) Comp. A- Comp. B

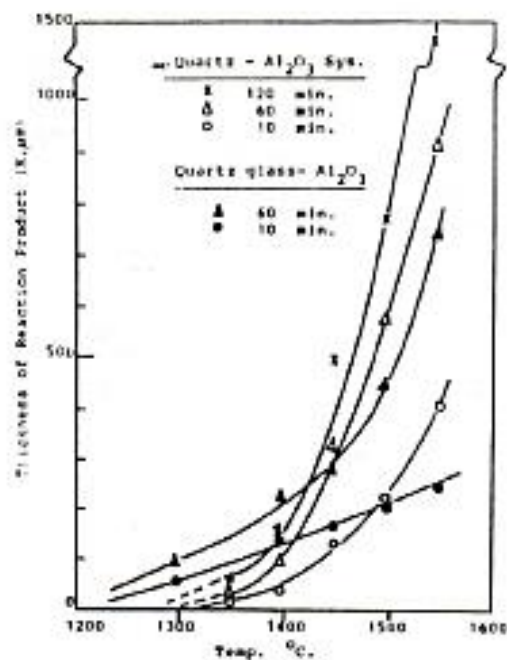


Fig. 2. Thickness of Reaction Product as a function of Temperature in  $\text{SiO}_2$ - $\text{Al}_2\text{O}_3$  System under Pressure.

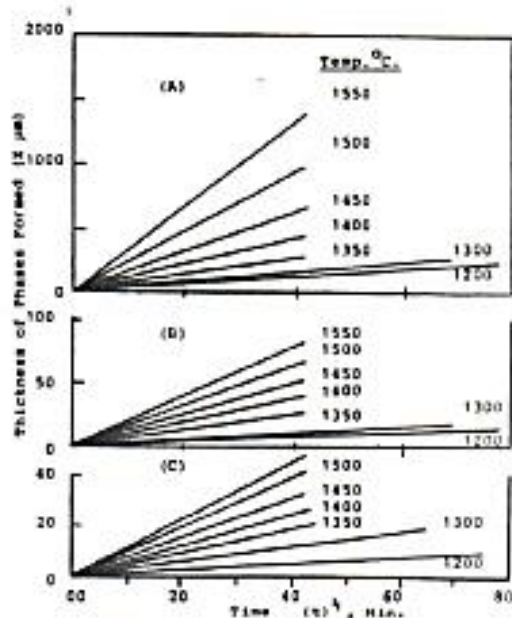


Fig. 3 Thickness of Phases Formed as a function of Time.

- (A) Mullite  $\text{A}_2\text{S}$       (B) Mullite  $\text{A}_3\text{S}_2$   
 (C) Glass Phases.



Due to the increase in volume, in case of quartz crystal, under constant pressure resulted large number of cracks in which  $Al_2O_3$  can easily diffuse and reacted with  $SiO_2$  giving more reaction interfaces. Consequently the rate of reaction ( $dR/dt$ ) increased than that in quartz glass compact. In both cases the reaction increased with time and temperature. The elemental analysis of the line scan obtained from the heat treated samples at 1200-1550°C, in the system quartz glass compact  $Al_2O_3$  powder showed the presence of mullite  $A_3S_2$  and  $A_2S$  and glass phase. Mullite  $A_2S$  is most probably formed inside  $SiO_2$  compact by solid state diffusion of  $Al_2O_3$  in  $SiO_2$  as was indicated by SEMPA. The formation of  $A_3S_2$  and glass was attributed to the solid state diffusion of  $SiO_2$  in  $Al_2O_3$ . The thickness of these phases was plotted against the reaction time ( $t$ ) at various temperatures as shown in Fig. 3. The straight lines obtained follow the relationship;

$$X = C t^{1/2}$$

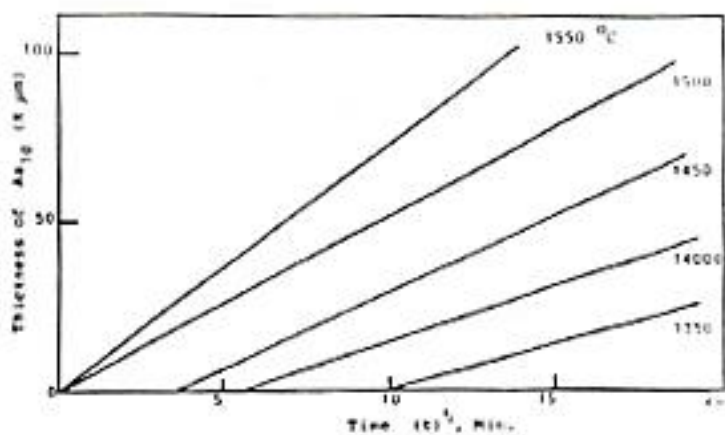
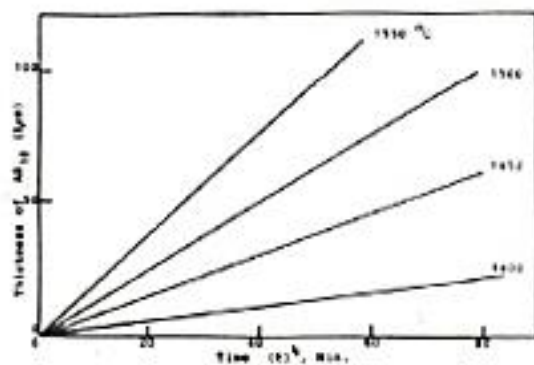
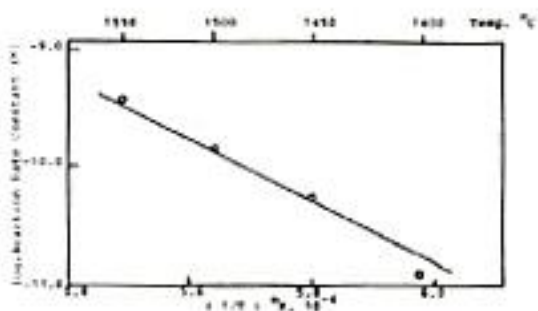
where

(X) is the thickness of reaction zone and (C) is constant. The ratio between  $A_2S$ :  $A_3S_2$ : glass is 100: 6: 4 below 1450°C while above it the ratio is 10: 1: 1. This shows clearly the influence of temperature on the rate of formation of different phases. Fig. 4 shows the photomicrograph of the reacted sample at 1400°C for 2 hours. It shows clearly the reaction zone between  $SiO_2$  compact and  $Al_2O_3$  powder. The relationship between the reaction rate constant, K, and the reciprocal of absolute temperature,  $1/T$ , is given in Fig. 5. The computed values of activation energy obtained, from Arrhenius plots, were 415, 340 and 210 KJ/mole for  $A_2S$ ,  $A_3S_2$  and glass. These reflects the easier formation of glass than mullite. This glass phase is not stable, depending on the rate of cooling of the reacted samples. This may be decomposed to mullite as an intermediate phase(5).

#### Reaction Between $\alpha-Al_2O_3$ Compact and $SiO_2$ Powder

Compact of  $\alpha-Al_2O_3$  (corundum or pressed  $\alpha-Al_2O_3$ ) was embedded in  $SiO_2$  powder, then hot pressed under 275 bar at 1200-1550°C for 10-360 minutes. The samples were examined at the reaction interface with SEMPA and light microscope. The measured thickness of the reaction product at different temperatures was plotted against the corresponding time of reaction as shown in Fig. 6. In this reaction system, the rate of reaction is very low compared to the reaction in quartz compact- $Al_2O_3$  powder system. On the other hand, the annealing of pre-hot pressed samples showed that below 1400°C no reaction between  $\alpha-Al_2O_3$  compact and  $SiO_2$  powder was detected even for long time of reaction (168 hrs.). At 1400°C, at the reaction interface only, one phase is present which is composed of 85%  $SiO_2$  and 15%  $Al_2O_3$  or  $AS_{10}$ . This phase may be formed by the solid state diffusion of  $Al_2O_3$  in  $SiO_2$  as was indicated from the lines of scan. The thickness of this phase is plotted against corresponding time of reaction at 1400-1550°C as shown in Fig. 7. The value of apparent activation energy obtained from Arrhenius plots Fig. 8, is 540 KJ/mole. This indicates that the formation of this phase is not as simple as the other glass obtained in the  $SiO_2$  compact-



Fig. 4 Thickness of Phase  $Al_{10}$  as a function of Time.Fig. 7 Thickness of  $Al_{10}$  as a function of Time at 1400-1550°C.Fig. 8. Arrhenius Plot for the Reaction  $W-Al_2O_3$  Compact- $SiO_2$  Powder.





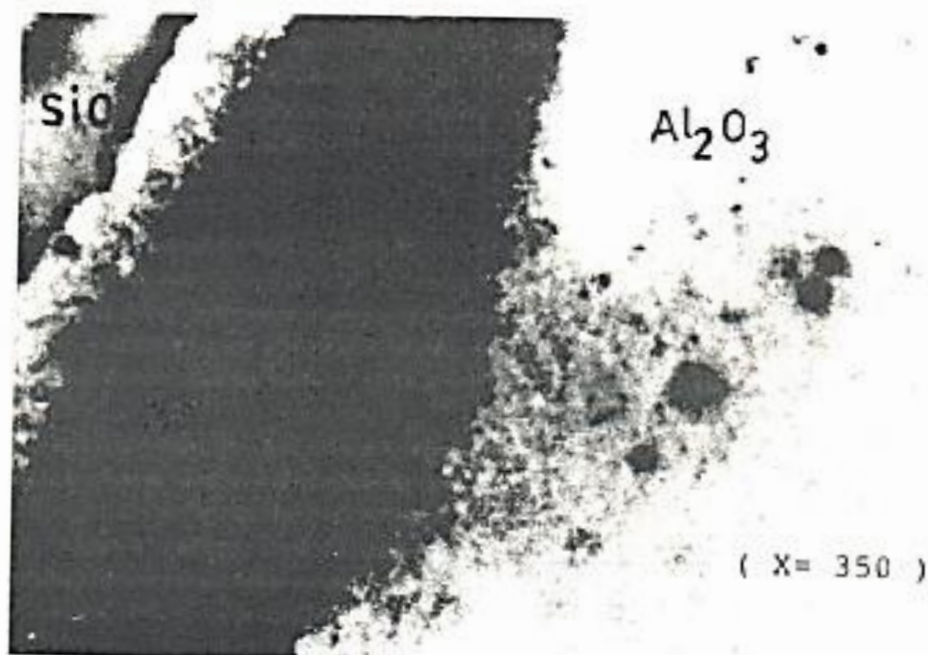


Fig. 9 Photomicrograph showing two Zones of Reaction between SiO<sub>2</sub> Powder and Al<sub>2</sub>O<sub>3</sub> Powder.

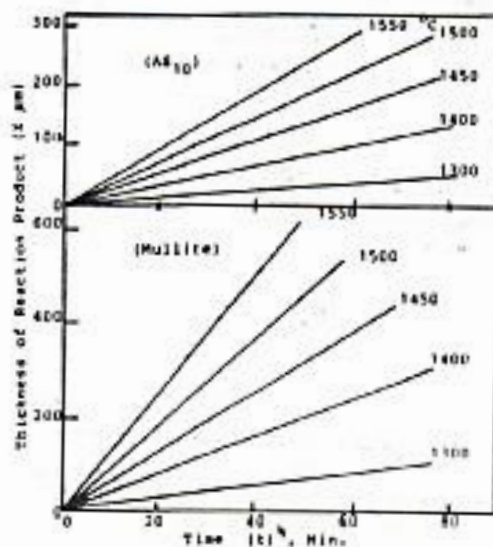


Fig. 10 Thickness of Phases Formed as a function of Time.



Table 3. Phases formed between  $\text{SiO}_2$  and  $\text{Al}_2\text{O}_3$ 

Reaction system	Reaction products	$\Delta E$ KJ/mole	Diffusion state
Compact $\text{SiO}_2$ - powder $\text{Al}_2\text{O}_3$	mullite $\text{A}_2\text{S}$	415	$\text{Al}_2\text{O}_3$ in $\text{SiO}_2$
	mullite $\text{A}_3\text{S}_2$	340	$\text{SiO}_2$ in $\text{Al}_2\text{O}_3$
	glasses	210	$\text{SiO}_2$ in $\text{Al}_2\text{O}_3$
Compact $\text{Al}_2\text{O}_3$ - powder $\text{SiO}_2$	glass $\text{AS}_{10}$	540	$\text{Al}_2\text{O}_3$ in $\text{SiO}_2$
Powder $\text{SiO}_2$ - Powder $\text{Al}_2\text{O}_3$	mullite $\text{A}_2\text{S}$	185	$\text{Al}_2\text{O}_3$ in $\text{SiO}_2$
	mullite $\text{A}_3\text{S}_2$		$\text{SiO}_2$ in $\text{Al}_2\text{O}_3$
	glasses		$\text{SiO}_2$ in $\text{Al}_2\text{O}_3$
	glass $\text{AS}_{10}$	350	$\text{Al}_2\text{O}_3$ in $\text{SiO}_2$
Compact $\text{SiO}_2$ - Compact $\text{Al}_2\text{O}_3$	glass $\text{AS}_{10}$	570	$\text{Al}_2\text{O}_3$ in $\text{SiO}_2$

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